

# Controlling Bulk Optical Properties of Emissive Polymersomes through Intramembranous Polymer–Fluorophore Interactions

P. Peter Ghoroghchian,<sup>†,‡</sup> Paul R. Frail,<sup>‡</sup> Guizhi Li,<sup>‡</sup> John A. Zupancich,<sup>§</sup> Frank S. Bates,<sup>§</sup> Daniel A. Hammer,<sup>\*,†</sup> and Michael J. Therien<sup>\*,‡</sup>

School of Engineering and Applied Science and Institute for Medicine and Engineering, University of Pennsylvania, 120 Hayden Hall, 3320 Smith Walk, Philadelphia, Pennsylvania 19104, Department of Chemistry, University of Pennsylvania, 231 South 34th Street, Philadelphia, Pennsylvania 19104, and Department of Chemical Engineering and Materials Science, University of Minnesota, 151 Amundson Hall, 421 Washington Avenue Southeast, Minneapolis, Minnesota 55455

Received October 11, 2006. Revised Manuscript Received November 23, 2006

Interdisciplinary investigation at the interface of chemistry, engineering, and medicine has enabled the development of self-assembled nanomaterials with novel biochemical and electro-optical properties. We have recently shown that emissive polymersomes, polymer vesicles incorporating porphyrin-based fluorophores, feature large integrated-emission oscillator strengths and narrow emission bands; these nanoscale assemblies can be further engineered to fluoresce at discrete wavelengths throughout the visible and near-infrared (NIR) spectral domains. As such, emissive polymersomes effectively define an organic-based family of soft-matter quantum-dot analogs that possess not only impressive optical properties, but also tunable physical and biomaterial characteristics relative to inorganic fluorescent nanoparticles. Here, we expand upon our initial studies on poly(ethyleneoxide)-block-poly(butadiene)-based vesicles to examine fluorophore membrane-loading in other polymersome systems. Through modulation of fluorophore ancillary group substituents and choice of polymer chain chemistries, we are able to predictably control intramembranous polymer–fluorophore interactions; these phenomena, in turn, influence the nature of fluorophore solvation, local dielectric environment, and emission quantum yield within emissive polymersome assemblies. By utilizing different classes of vesicle-generating diblock copolymers, including bioresorbable poly(ethyleneoxide)-block-poly( $\epsilon$ -caprolactone) (PEO-b-PCL) and poly(ethyleneoxide)-block-poly( $\gamma$ -methyl- $\epsilon$ -caprolactone) (PEO-b-PMCL), we ascertain general principles important for engineering nanoscale optical vesicles. Further, this work heralds the first generation of fully biodegradable fluorescent nanoparticles suitable for deep-tissue *in vivo* imaging.

## Introduction

Although conventional visible fluorophores are routinely utilized for *in vitro* biological research, there has been increasing interest in the development of near-infrared (NIR) emissive agents suitable for deep-tissue *in vivo* imaging.<sup>1–6</sup> To date, many investigators have focused on the development of quantum dots (inorganic semiconductor nanocrystals) for both *in vitro* and *in vivo* applications.<sup>7–11</sup> Although quantum dots exhibit vastly superior optical characteristics when

compared to traditional organic fluorophores,<sup>12,13</sup> they may be of limited clinical utility, because their metallic composites are inherently toxic.<sup>14,15</sup> As such, the design of alternative fluorescent nanoparticles is an area of active investigation and aims to generate novel organic-based agents that possess large-magnitude irradiance and high photostability.

Meso-to-meso ethyne-bridged oligo(porphinato)zinc(II)-based supermolecules (**PZn<sub>n</sub>** compounds) define a family of near-infrared fluorophores (NIRFs) that exhibit high molar absorptivities throughout the visible and NIR regions,<sup>16–19</sup>

\* To whom correspondence should be addressed. E-mail: hammer@seas.upenn.edu (D.A.H.); therien@sas.upenn.edu (M.J.T.).

<sup>†</sup> School of Engineering and Applied Science and Institute for Medicine and Engineering, University of Pennsylvania.

<sup>‡</sup> Department of Chemistry, University of Pennsylvania.

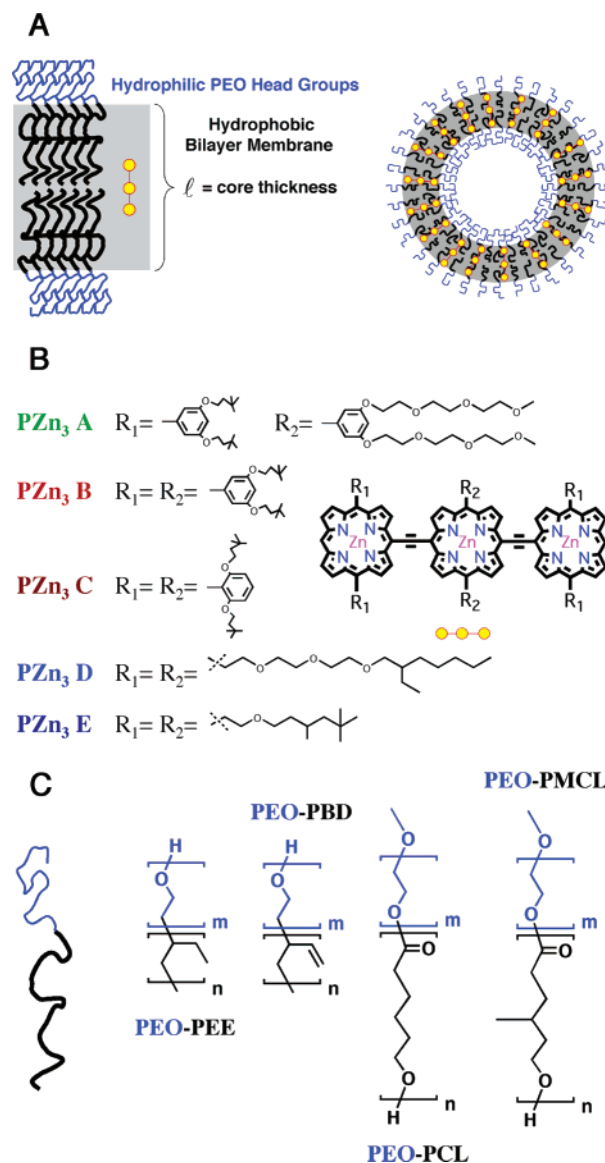
<sup>§</sup> Department of Chemical Engineering and Materials Science, University of Minnesota.

- (1) Weissleder, R.; Tung, C. H.; Mahmood, U.; Bogdanov, A., Jr. *Nat. Biotechnol.* **1999**, *17*, 375–378.
- (2) Achilefu, S.; Dorshow, R. B.; Bugaj, J. E.; Rajagopalan, R. *Invest. Radiol.* **2000**, *35*, 479–485.
- (3) Becker, A.; Hennesius, C.; Licha, K.; Ebert, B.; Sukowski, U.; Semmler, W.; Wiedenmann, B.; Grotzinger, C. *Nat. Biotechnol.* **2001**, *19*, 327–331.
- (4) Sevick-Muraca, E. M.; Houston, J. P.; Gurfinkel, M. *Curr. Opin. Chem. Biol.* **2002**, *6*, 642–650.
- (5) Frangioni, J. V. *Curr. Opin. Chem. Biol.* **2003**, *7*, 626–634.
- (6) Weissleder, R.; Ntziachristos, V. *Nat. Med.* **2003**, *9*, 123–128.
- (7) Larson, D. R.; Zipfel, W. R.; Williams, R. M.; Clark, S. W.; Bruchez, M. P.; Wise, F. W.; Webb, W. W. *Science* **2003**, *300*, 1434–1436.

- (8) Wu, X. Y.; Liu, H. J.; Liu, J. Q.; Haley, K. N.; Treadway, J. A.; Larson, J. P.; Ge, N. F.; Peale, F.; Bruchez, M. P. *Nat. Biotechnol.* **2003**, *21*, 41–46.
- (9) Voura, E. B.; Jaiswal, J. K.; Mattoussi, H.; Simon, S. M. *Nat. Med.* **2004**, *10*, 993–998.
- (10) Kim, S.; Lim, Y. T.; Soltész, E. G.; De Grand, A. M.; Lee, J.; Nakayama, A.; Parker, J. A.; Mihaljevic, T.; Laurence, R. G.; Dor, D. M.; Cohn, L. H.; Bawendi, M. G.; Frangioni, J. V. *Nat. Biotechnol.* **2004**, *22*, 93–97.
- (11) Gao, X. H.; Cui, Y. Y.; Levenson, R. M.; Chung, L. W. K.; Nie, S. M. *Nat. Biotechnol.* **2004**, *22*, 969–976.
- (12) Dabbousi, B. O.; Rodriguez-Viejo, J.; Mikulec, F. V.; Heine, J. R.; Mattoussi, H.; Ober, R.; Jensen, K. F.; Bawendi, M. G. *J. Phys. Chem. B* **1997**, *101*, 9463–9475.
- (13) Chan, W. C. W.; Nie, S. M. *Science* **1998**, *281*, 2016–2018.
- (14) Jaiswal, J. K.; Simon, S. M. *Trends Cell Biol.* **2004**, *14*, 497–504.
- (15) Frangioni, J. V. *Nat. Biotechnol.* **2006**, *24*, 326–328.
- (16) Lin, V. S.-Y.; DiMaggio, S. G.; Therien, M. J. *Science* **1994**, *264*, 1105–1111.

large NIR fluorescence quantum yields,<sup>20,21</sup> high photobleaching thresholds,<sup>21,22</sup> and no chemical- or photo-based in vitro toxicity.<sup>23</sup> Through cooperative self-assembly with amphiphilic diblock copolymers, these electronically conjugated porphyrinic NIRFs can be dispersed, noncovalently, at high concentrations (>10 mol/wt %) within the thick synthetic membranes of polymersomes<sup>24–26</sup> (50 nm to 50  $\mu$ m diameter polymer vesicles).<sup>27–29</sup> These NIR-emissive polymersomes define a family of organic-based, soft matter quantum dot analogues that are ideally suited for in vivo optical imaging.<sup>27</sup> Previous studies have focused upon the delineation of a rich photophysical diversity,<sup>28</sup> and quantification of fluorophore membrane-loading,<sup>29</sup> in these emissive assemblies; membrane incorporation of a wide range of related multiporphyrinic fluorophores has enabled precise emission energy modulation over a broad domain of the visible and near-infrared spectrum (600–900 nm).<sup>27,28</sup> Here, we expand upon our initial studies involving poly(ethyleneoxide)-block-poly(butadiene)-based polymersomes that disperse **PZn<sub>n</sub>** fluorophores, to develop a generalized methodology for engineering the bulk optical properties of other emissive vesicle assemblies. Through rational design of fluorophore ancillary group substituents and choice of polymer chain chemistries, we show that intramembranous polymer–fluorophore interactions can be extensively modulated to influence the nature of fluorophore solvation, mean electronic environment, and thus, the **PZn<sub>n</sub>** emission quantum yield.

We focus herein on the meso-to-meso ethynyl-bridged tris[porphyrinato]zinc(II)-based (**PZn<sub>3</sub>**) fluorophore and examine the effects of substituent-driven **PZn<sub>3</sub>**-dispersion in various polymersome environments, differing in both membrane core thickness ( $l$ ) and chemical composition (Figure 1), via steady-state absorption and emission spectroscopies. Five related **PZn<sub>3</sub>**-based fluorophores that differ with respect to ancillary group substituents (**PZn<sub>3</sub> A–E**, Figure 1B) were utilized. By comparing each vesicular nanoscale compositions' integrated steady-state emission intensity as a function of membrane-loading concentration, we elucidated the role of polymer chain/**PZn<sub>3</sub>** fluorophore local solvation effects in



**Figure 1.** Schematic of various meso-to-meso ethynyl-bridged tris[porphyrinato]zinc(II)-based (**PZn<sub>3</sub>**) fluorophores incorporated within different polymersome environments. (A) Illustration of **PZn<sub>3</sub>** dispersion within the vesicle's hydrophobic bilayer membrane. (B) Chemical structures of five **PZn<sub>3</sub>**-based fluorophores (**PZn<sub>3</sub> A–E**) differing in ancillary pendent-group substituents. (C) Chemical structures of biocompatible (PEO-b-PBD and PEO-b-PEE) and biodegradable (PEO-b-PCL and PEO-b-PMCL) vesicle-generating diblock copolymers.

determining the bulk optical properties of emissive polymersomal assemblies.

Several different classes of vesicle-generating diblock copolymers were utilized (Figure 1C), including biocompatible poly(ethyleneoxide)-block-poly(butadiene) (PEO-b-PBD) and poly(ethyleneoxide)-block-poly(ethylene) (PEO-b-PEE),<sup>24,30</sup> as well as biodegradable poly(ethyleneoxide)-block-poly( $\epsilon$ -caprolactone) (PEO-b-PCL) and poly(ethyleneoxide)-block-poly( $\gamma$ -methyl- $\epsilon$ -caprolactone) (PEO-b-PMCL) compositions.<sup>31,32</sup> Membrane loading of these

- (17) Lin, V. S.-Y.; Therien, M. J. *Chem.—Eur. J.* **1995**, *1*, 645–651.  
 (18) Angiolillo, P. J.; Lin, V. S.-Y.; Vanderkooi, J. M.; Therien, M. J. *J. Am. Chem. Soc.* **1995**, *117*, 12514–12527.  
 (19) Susumu, K.; Therien, M. J. *J. Am. Chem. Soc.* **2002**, *124*, 8550–8552.  
 (20) Rubtsov, I. V.; Susumu, K.; Rubtsov, G. I.; Therien, M. J. *J. Am. Chem. Soc.* **2003**, *125*, 2687–2696.  
 (21) Duncan, T. V.; Susumu, K.; Sinks, L. E.; Therien, M. J. *J. Am. Chem. Soc.* **2006**, *128*, 9000–9001.  
 (22) Susumu, K.; Frail, P. R.; Angiolillo, P. J.; Therien, M. J. *J. Am. Chem. Soc.* **2006**, *128*, 8380–8381.  
 (23) Wu, S. P.; Lee, I.; Ghoroghchian, P. P.; Frail, P. R.; Zheng, G.; Glickson, J. D.; Therien, M. J. *Bioconjugate Chem.* **2005**, *16*, 542–550.  
 (24) Discher, B. M.; Won, Y. Y.; Ege, D. S.; Lee, J. C. M.; Bates, F. S.; Discher, D. E.; Hammer, D. A. *Science* **1999**, *284*, 1143–1146.  
 (25) Discher, D. E.; Eisenberg, A. *Science* **2002**, *297*, 967–973.  
 (26) Antonietti, M.; Forster, S. *Adv. Mater.* **2003**, *15*, 1323–1333.  
 (27) Ghoroghchian, P. P.; Frail, P. R.; Susumu, K.; Blessington, D.; Brannan, A. K.; Bates, F. S.; Chance, B.; Hammer, D. A.; Therien, M. J. *Proc. Natl. Acad. Sci. U.S.A.* **2005**, *102*, 2922–2927.  
 (28) Ghoroghchian, P. P.; Frail, P. R.; Susumu, K.; Park, T. H.; Wu, S. P.; Uyeda, H. T.; Hammer, D. A.; Therien, M. J. *J. Am. Chem. Soc.* **2005**, *127*, 15388–15390.  
 (29) Ghoroghchian, P. P.; Lin, J. J.; Brannan, A. K.; Frail, P. R.; Bates, F. S.; Therien, M. J.; Hammer, D. A. *Soft Matter* **2006**, *2*, 973–980.

- (30) Bermudez, H.; Brannan, A. K.; Hammer, D. A.; Bates, F. S.; Discher, D. E. *Macromolecules* **2002**, *35*, 8203–8208.  
 (31) Ghoroghchian, P. P.; Li, G.; Levine, D. H.; Davis, K. P.; Bates, F. S.; Hammer, D. A.; Therien, M. J. *Macromolecules* **2006**, *39*, 1673–1675.  
 (32) Zupancich, J. A.; Bates, F. S.; Hillmyer, M. A. *Macromolecules* **2006**, *39*, 4286–4288.

**PZn<sub>3</sub>** fluorophores in PEO<sub>30</sub>-b-PBD<sub>46</sub> ( $\ell = 9.6$  nm) vs PEO<sub>80</sub>-b-PBD<sub>125</sub> ( $\ell = 14.8$  nm)-based vesicles was compared in order to assess the effect of increasing the membrane core thickness on relative fluorophore distribution and intermolecular spacing, while maintaining a uniform **PZn<sub>3</sub>** solvation environment. Similar experiments comparing **PZn<sub>3</sub>** loading in PEO<sub>30</sub>-b-PBD<sub>46</sub>- and PEO<sub>40</sub>-b-PEE<sub>37</sub>-based vesicles were conducted to ascertain the roles that local solvation and interchain packing effects play in influencing the extent of fluorophore–fluorophore  $\pi$ – $\pi$  interactions and the degree to which the emitter is dispersed in the bilayer. Finally, membrane-loading studies of **PZn<sub>3</sub>**-based fluorophores in two biodegradable vesicle systems (PEO<sub>45</sub>-b-PCL<sub>105</sub> and PEO<sub>43</sub>-b-PMCL<sub>66</sub>) were performed to evaluate how membrane crystallinity influences **PZn<sub>3</sub>** solvation and mean electronic environment.

## Materials and Methods

**Materials.** PEO<sub>30</sub>-b-PBD<sub>46</sub> diblock copolymers under study were purchased from Polymer Source Inc. (Dorval, Quebec). PEO<sub>80</sub>-b-PBD<sub>125</sub>, PEO<sub>40</sub>-b-PEE<sub>37</sub>, PEO<sub>45</sub>-b-PCL<sub>105</sub>, and PEO<sub>43</sub>-b-PMCL<sub>66</sub> diblock polymers<sup>30–32</sup> as well as **PZn<sub>3</sub>** NIRF species **A–C**<sup>19,23</sup> were synthesized following procedures described previously.

**5,15-Bis[[5',-10',20'-bis([11-ethyl-3,6,9-trioxapentadecane-1-yl]porphinato)zinc(II)ethynyl]-10,20-bis([11-ethyl-3,6,9-trioxapentadecane-1-yl]porphinato)zinc(II)] (PZn<sub>3</sub> D).** (5-Ethynyl-10,20-bis(11-ethyl-3,6,9-trioxapentadecane-1-yl)porphinato)zinc(II) (46.2 mg,  $5.21 \times 10^{-5}$  mol) and (5,15-dibromo-10,20-bis(11-ethyl-3,6,9-trioxapentadecane-1-yl)porphinato)zinc(II) (24.2 mg,  $2.37 \times 10^{-5}$  mol) were charged into a 100 mL Schlenk tube along with Pd<sub>2</sub>-dba<sub>3</sub> (3 mg,  $3.55 \times 10^{-6}$  mol) and AsPh<sub>3</sub> (10 mg,  $2.84 \times 10^{-5}$  mol). A previously degassed solution of THF:TEA (9:1) was cannula-transferred into the reaction flask; the reaction mixture was then heated to 60 °C and stirred overnight under Ar. The reaction was quenched with water, diluted with CHCl<sub>3</sub>, and washed three times with aq. NH<sub>4</sub>Cl. The organic layer was collected, dried with Na<sub>2</sub>SO<sub>4</sub>, filtered, and evaporated. The residue was purified by flash chromatography on silica using CHCl<sub>3</sub>:pyridine (99:1) as the eluent, followed by gravimetric size-exclusion chromatography (Biobeads, SX-1, THF). The fastest moving band was collected. The product was purified a final time by flash chromatography on silica. Yield = 48 mg (77 mol %, based on (5,15-dibromo-10,20-bis(11-ethyl-3,6,9-trioxapentadecane-1-yl)porphinato)zinc(II)). <sup>1</sup>H NMR (500 MHz, CDCl<sub>3</sub>; 1 drop pyridine-d<sub>5</sub>):  $\delta$  10.56 (d, 4H,  $J = 4.45$  Hz), 10.50 (d, 4H,  $J = 4.45$  Hz), 9.98 (s, 2H), 9.87 (d, 4H,  $J = 4.50$  Hz), 9.80 (d, 4H,  $J = 4.50$  Hz), 9.66 (d, 4H,  $J = 4.40$  Hz), 9.33 (d, 4H,  $J = 4.35$  Hz), 5.12 (t, 4H,  $J = 7.50$  Hz), 4.37 (t, 4H,  $J = 7.26$  Hz), 4.14 (s, 2H), 3.55 (t, 4H,  $J = 4.01$  Hz), 3.46 (t, 4H,  $J = 4.84$  Hz), 3.08 (t, 4H,  $J = 4.63$  Hz), 2.89 (t, 4H,  $J = 4.64$  Hz), 1.50 (m, 42H), 1.28 (m, 69H), 0.89 (m, H). Vis (THF)  $\lambda_{\max}$  (log  $\epsilon$ ): 414 (5.17), 494 (5.42), 574 (4.17), 774 (5.03) nm. MALDI-TOF MS  $m/z$ : 2623.3485 (M<sup>+</sup>) (calculated for C<sub>148</sub>H<sub>200</sub>N<sub>12</sub>O<sub>18</sub>Zn<sub>3</sub>, 2625.2978).

**5,15-Bis[[5',-10',20'-bis([3-(3',5',5'-trimethylhexyloxy)-ethyl]porphinato)zinc(II)ethynyl]-10,20-bis([3-(3',5',5'-trimethylhexyloxy)-ethyl]porphinato)zinc(II)] (PZn<sub>3</sub> E).** (5,15-Dibromo-10,20-bis(3-(3',5',5'-trimethylhexyloxy)-ethyl)porphinato)zinc(II) (71 mg,  $8.13 \times 10^{-5}$  mol) and (5-ethynyl-10,20-bis(3-(3',5',5'-trimethylhexyloxy)-ethyl)porphinato)zinc(II) (132 mg,  $1.79 \times 10^{-4}$  mol) were charged into a 100 mL Schlenk tube along with Pd<sub>2</sub>-dba<sub>3</sub> (25 mg,  $2.69 \times 10^{-5}$  mol) and AsPh<sub>3</sub> (70 mg,  $2.15 \times 10^{-4}$  mol). A previously degassed solution of THF:TEA (9:1) was cannula-transferred into the reaction flask; the reaction mixture was then

heated to 60 °C and stirred overnight under Ar. The reaction was quenched with water, diluted with CHCl<sub>3</sub>, and washed three times with aq. NH<sub>4</sub>Cl. The organic layer was collected, dried with Na<sub>2</sub>SO<sub>4</sub>, filtered, and evaporated. The residue was purified by flash chromatography on silica using CHCl<sub>3</sub>:pyridine (99:1) as the eluent, followed by gravimetric size-exclusion chromatography (Biobeads, SX-1, THF). The fastest moving band was collected. The product was purified a final time by flash chromatography on silica. Yield = 137 mg (77 mol %, based on (5,15-dibromo-10,20-bis(3-(3',5',5'-trimethylhexyloxy)-ethyl)porphinato)zinc(II)). <sup>1</sup>H NMR (500 MHz, CDCl<sub>3</sub>):  $\delta$  10.56 (d, 4H,  $J = 4.15$  Hz), 10.49 (d, 4H,  $J = 4.05$  Hz), 9.96 (s, 2H), 9.84 (d, 4H,  $J = 0.385$  Hz), 9.77 (d, 4H,  $J = 4.45$  Hz), 9.63 (d, 4H,  $J = 4.30$  Hz), 9.32 (d, 4H,  $J = 4.10$  Hz), 5.38 (br-m, 12H), 4.63 (t, 12H,  $J = 8.05$  Hz), 4.57 (t, 12H,  $J = 8.06$  Hz), 3.74 (t, 12H,  $J = 6.55$  Hz), 3.69 (t, 12H,  $J = 5.47$  Hz), 1.62 (m, 12H), 1.45 (m, 12H), 1.30 (m, 12H), 1.19 (m, 12H), 1.09 (m, 6H), 0.87 (m, 54H). Vis (THF)  $\lambda_{\max}$  (log  $\epsilon$ ): 414.00 (5.17), 494 (5.42), 574 (4.22), 774 (5.03) nm. MALDI-TOF MS  $m/z$ : 2182.9642 (M<sup>+</sup>) (calculated for C<sub>130</sub>H<sub>164</sub>N<sub>12</sub>O<sub>6</sub>Zn<sub>3</sub>, 2181.0771).

**Vesicle Preparation.** Formation of small (<300 nm diameter) emissive polymersomes followed procedures described previously.<sup>28</sup> Briefly, 1 mM diblock copolymer solutions and 1 mM solutions of NIRF species **PZn<sub>3</sub> A–E** in 10% THF:CH<sub>2</sub>Cl<sub>2</sub> were combined in various molar ratios and uniformly coated on the surfaces of roughened Teflon plates, followed by evaporation of the solvent under vacuum for >12 h. The Teflon plate consisted of a 1/16 in. thick sheet cut in 1 in. by 1 in. squares and machine-roughened with industrial grade sandpaper (60 grit, aluminum oxide) to create macroscopically uniform digitations. Each dry Teflon plate containing the polymer:fluorophore sample was placed in a 20 mL glass vial; 2 mL of aqueous solution (e.g., DI water) were then added. The vial was subsequently sealed, immersed in a bath sonicator, and heated at 60 °C for 1 h, leading to spontaneous budding of medium sized (~300 nm diameter) emissive polymersomes (membrane loaded at the prescribed polymer:fluorophore molar ratio) off of the Teflon and into the aqueous surroundings. Small unilamellar polymersomes that possess appropriately narrow size distributions were prepared via procedures analogous to those used to formulate small lipid vesicles (sonication, freeze–thaw extraction, and extrusion).<sup>27</sup>

**Steady-State Electronic Absorption Spectroscopy of Emissive Polymersomes that Disperse PZn<sub>3</sub> A–E Fluorophores.** Aqueous suspensions (in DI water) of small, unilamellar NIR-emissive polymersomes (loaded at a 20:1 polymer:NIRF molar ratio;  $n = 5$  for each unique polymer:fluorophore formulation) were placed in 10 mm quartz optical cells; electronic absorption spectra for each of the membrane-incorporated fluorophore species were recorded using an OLIS UV/vis/NIR spectrophotometry system that is based on the optics of a Cary 14 spectrophotometer (On-line Instrument Systems Inc, Bogart, GA). The solutions were then transferred to a glass vial, frozen in liquid N<sub>2</sub>, and lyophilized (FreeZone 4.5 L benchtop freeze dry system, Labconco Corporation, Kansas City, MO; model 77500) for 24 h to destroy the vesicles and dry the polymer and fluorophore species. The dry samples were then taken up in the same volume of THF and their absorption spectra were recorded. NIRF concentrations in the original polymersome solutions were calculated via Beer's Law using the THF absorption spectra and the previously determined molar extinction coefficients ( $\epsilon$ , M<sup>-1</sup> cm<sup>-1</sup>) for each emitter in this solvent. These calculated concentrations, as well as the original NIRF absorption spectra in aqueous polymersome solutions, were then utilized in order to determine the  $\epsilon$  values for each polymersome-loaded fluorophore.

**Steady-State Fluorescence Spectroscopy of Emissive Polymersomes that Disperse PZn<sub>3</sub> A–E Fluorophores.** Aqueous

suspensions (in DI water) of small, unilamellar NIR-emissive polymersomes (loaded at a 20:1 polymer:NIRF molar ratio;  $n = 5$  for each unique polymer:fluorophore formulation) were placed in 10 mm quartz optical cells and steady-state emission spectra were obtained using a Spex Fluorolog-3 spectrophotometer (Jobin Yvon Inc, Edison, NJ) that utilized a dual S and T channel configuration and PMT/InGaAs/Extended-InGaAs detectors ( $\lambda_{\text{ex}} = 510 \text{ nm}$ ). Sample concentrations were modulated so that the optical density at the excitation wavelength was  $<0.05$  O.D. All spectra were collected with the single excitation and emission apertures set at 5 nm. Fluorescence spectra were corrected to account for the wavelength-dependent efficiency of the detection system that was determined using the spectral output of a calibrated light source obtained from the National Bureau of Standards.

**Comparison of Relative Concentration-Dependent Fluorescence Intensities of  $\text{PZn}_3$  A and B.** Suspensions (in DI water) of small ( $<300 \text{ nm}$  diameter)  $\text{PEO}_{30}\text{-b-PBD}_{46}$ -,  $\text{PEO}_{80}\text{-b-PBD}_{125}$ -,  $\text{PEO}_{40}\text{-b-PEE}_{37}$ -,  $\text{PEO}_{45}\text{-b-PCL}_{105}$ -, and  $\text{PEO}_{43}\text{-b-PMCL}_{66}$ -based emissive polymersomes were generated by combining the respective diblock copolymers with NIRFs  $\text{PZn}_3$  A and B in various molar ratios of polymer:NIRF (500:1, 250:1, 100:1, 40:1, 20:1, 10:1, 5:1; note,  $n = 3$  for each unique polymer:NIRF combination). Steady-state absorption spectra for each of the solutions were recorded, and 250  $\mu\text{L}$  of each sample was transferred to black 96-well plates ( $n = 3$  plates; Becton Dickinson Co.).

Near-infrared fluorescence from each well in the plate was then detected and quantified with an eXplore Optix imaging system (GE Healthcare). The instrument used the following experimental parameters: 785 nm excitation laser,  $785 \pm 5 \text{ nm}$  band-pass excitation filter,  $850 \pm 50 \text{ nm}$  band-pass emission filter, 0.1 s integration time, 1 mm step size, and  $25 \text{ }^\circ\text{C}$  stage temperature. The fluorescence intensity from each solution well in the plate was normalized by the laser power using analysis software provided by the instrument manufacturer (Analysis WorkStation 1.1.3.0, Advanced Research Technologies, Quebec, Canada). Finally, the normalized integrated-emitted-photon counts from each sample well were divided by the NIRF concentration at the excitation wavelength (determined from the steady-state absorption spectra obtained for each emissive polymersome sample), compared, and divided by the largest sample value. The average values for normalized emission from  $\text{PZn}_3$  A and B are plotted against the calculated fluorophore concentration in the polymersome membrane and reported in Figures 4–6; note, the individual values determined for each sample in a plate varied by less than 10% from the reported average-normalized-emission intensity for each unique emissive polymersome formulation.

## Results and Discussion

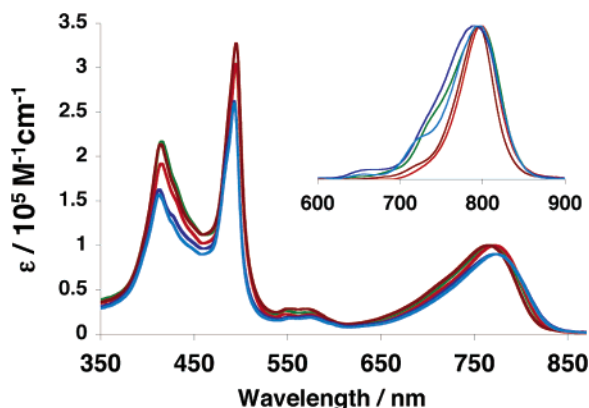
**Design of  $\text{PZn}_3$  A–E.** We have previously demonstrated that meso-to-meso ethynyl-bridged tris[porphinato]zinc(II) ( $\text{PZn}_3$ ) fluorophores possess not only strong near-infrared absorption bands ( $\epsilon > 100\,000$  at  $\lambda = 767 \text{ nm}$ )<sup>16,19</sup> but also large NIR fluorescence quantum yields ( $\lambda_{\text{em,max}} = 806 \text{ nm}$ ,  $\phi_{\text{F}} = 22 \pm 3\%$ )<sup>21</sup> in tetrahydrofuran (THF) solvent. Moreover, these hydrophobic  $\text{PZn}_3$ -based fluorophores are readily dispersed in aqueous solution when incorporated within the thick membranes of synthetic polymer vesicles. Emissive polymersomes, dispersing PZn-based fluorophores at precise membrane concentrations, are self-assembled through a simple and quantitative method involving aqueous hydration of dry, uniform thin films of amphiphilic polymer and fluorophore deposited in prescribed molar ratios on rough-

ened Teflon surfaces (see the Experimental Section).<sup>27,29</sup> Notably, our earlier work determined that  $\text{PZn}_3$  ancillary aryl group substituents influence not only interplanar torsional angle distribution of fluorophore component PZn macrocycles, and thus the magnitude of PZn-to-PZn conjugative interactions,<sup>28</sup> but also the relative fluorophore distribution within the hydrophobic bilayer of PEO-b-PBD-based vesicles. As a result, we focused our current efforts on designing several new  $\text{PZn}_3$ -based fluorophores with differing pendent ancillary groups in order to identify key intramembraneous molecular interactions that would enable optimization of polymersome bulk optical properties.

Although  $\text{PZn}_3$  A–E each possess an identical conjugated chromophoric core, their ancillary pendent groups vary (Figure 1), impacting the nature of the molecular interactions of these fluorophores with the polymer chains composing the vesicles' membranes. For example, we have previously shown that modulating the amphiphilicity of these PZn-based fluorophores affords control over their optical properties within PEO-b-PBD-based polymersomes.<sup>28</sup> Although both  $\text{PZn}_3$  A and B possess 3',5'-disubstituted pendent aryl groups, replacement of  $\text{PZn}_3$  A's 9-methoxy-1,4,7-trioxanonyl substituents with 3,3-dimethyl-1-butyloxy units augments the fluorophore's hydrophobicity. As a result,  $\text{PZn}_3$  B's relative membrane localization is shifted away from the environment defined by the interface of the hydrophilic PEO and hydrophobic PBD block fractions (location of  $\text{PZn}_3$  A) and facilitates its predominant dispersion into the hydrophobic core of the PBD membrane.<sup>28</sup> Further, varying aryl group substitution from a 3',5'-( $\text{PZn}_3$  B) to a 2',6'-substitution pattern ( $\text{PZn}_3$  C) converts overall chromophore geometry from a biconcave wedge to a cylinder and influences the local arrangement of surrounding polymer chains within the vesicle membrane.<sup>28</sup>

Given these earlier observations with PEO-b-PBD-based emissive polymersomes, we synthesized two new  $\text{PZn}_3$  fluorophores ( $\text{PZn}_3$  D and E) that lack 10- and 20-pendent aryl groups; these species possess, respectively, 11-ethyl-3,6,9-trioxapentadecane-1-yl ( $\text{PZn}_3$  D) and 3-(3',5',5'-trimethylhexyloxy)-ethyl ( $\text{PZn}_3$  E) ancillary substituents, directly linked to the  $\text{PZn}_3$  component macrocycle 10- and 20-positions. These design variations were intended to potentially augment fluorophore–fluorophore intermolecular spacing by minimizing both the disruption of polymer chain packing and the required  $\text{PZn}_3$ -dissolution volume within the vesicle membrane. Likewise, the differences between  $\text{PZn}_3$  D and E's ancillary substituent groups were expected to affect their relative membrane dispersion in a fashion similar to that evinced in earlier experiments involving the comparison of  $\text{PZn}_3$  A and B.<sup>28</sup>

**Electronic Absorption Spectra of  $\text{PZn}_3$  A–E in Various Polymersome Environments.** Although  $\text{PZn}_3$ -based fluorophores A–E differ with respect to the nature and location of their ancillary group substituents, they display nearly identical absorptive and emissive spectral characteristics in THF solvent (Figure 2). The same fluorophores, however, show marked spectral differences depending on the polymersome environment in which they are noncovalently incorporated (Figure 3). When comparing the absorptive and



**Figure 2.** Steady-state electronic absorption and emission (inset) spectra of **PZn<sub>3</sub>**-based fluorophores **A–E** in tetrahydrofuran solvent. Compound color scheme corresponds to that displayed in Figure 1B;  $\lambda_{\text{ex}} = 510$  nm.

emissive signatures of **PZn<sub>3</sub>** **A–E** in aqueous suspensions of PEO-b-PBD (Figure 3A) and PEO-b-PEE-based (Figure 3B) vesicles to their THF solution spectra (Figure 2), it is readily apparent that ancillary group substituents alter the nature of **PZn<sub>3</sub>** dissolution in a polymersome-specific manner. For example, although **PZn<sub>3</sub>**-based fluorophores **D** (light blue) and **E** (dark blue), which lack pendent aryl groups, display spectral features consistent with extensive fluorophore aggregation within PBD-based membranes (Figure 3A), they are more effectively dispersed in the fully saturated PEE-based bilayers (Figure 3B). In particular, **PZn<sub>3</sub>** **D** shows a significant decrease in spectral heterogeneity,<sup>19–21,28</sup> signaling a narrower PZn–PZn torsional angle distribution about a diminished mean torsional angle, within these PEO-b-PEE-based vesicles (Figure 3B) relative to that manifest within other polymersome environments (Figure 3A, C, and D).

As seen in Figure 3B, relative dissolution of **PZn<sub>3</sub>** **A** (green), **B** (red), and **C** (crimson) within PEE-based membranes is similar to that observed previously for the same fluorophores dispersed in PBD (Figure 3A). Namely, the relative proportion of ancillary 3,3-dimethyl-1-butyloxy and 9-methoxy-1,4,7-trioxanonyl aryl-substituents affects the emission energy, as well as the spectral breadth of the low-energy absorptive and emissive transitions, of these supermolecular fluorophores. Hydrophobic 3,3-dimethyl-1-butyloxy groups preferentially disperse **PZn<sub>3</sub>**-based fluorophores into the core of the bilayer membrane (**PZn<sub>3</sub>** **B**), whereas increasing numbers of 9-methoxy-1,4,7-trioxanonyl groups enhance the extent to which these chromophores are dispersed at the membrane environment defined by the interface of the hydrophilic (PEO) and hydrophobic (PBD or PEE) block fractions (**PZn<sub>3</sub>** **A**).<sup>28,29</sup>

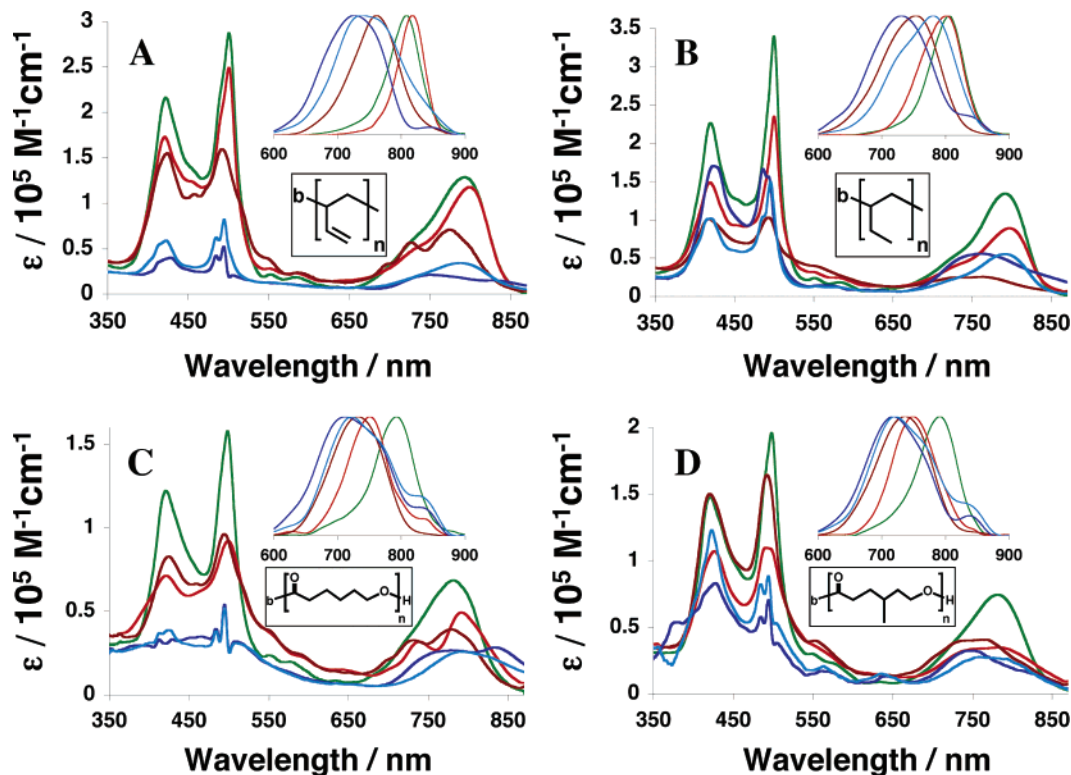
In dilute THF solution, the position of the ancillary aryl 3,3-dimethyl-1-butyloxy substituent does not impact fluorophore optical properties: chromophores having 2',6'-disubstituted aryl rings (**PZn<sub>3</sub>** **C**, Figure 2 crimson) exhibit the same absorptive and emissive signatures as those that feature 3',5'-disubstituted aryl groups (**PZn<sub>3</sub>** **B**, Figure 2, red). By comparison, however, within PEO-b-PEE and PEO-b-PBD-based polymersome environments, chromophores bearing 2',6'-di(3,3-dimethyl-1-butyloxy)phenyl rings display fluorescence band hypsochromic shifts (**PZn<sub>3</sub>** **C**, panels A

and B of Figure 3, crimson), whereas the same chromophores possessing a 3',5'-substitution pattern exhibit bathochromically shifted emission bands (**PZn<sub>3</sub>** **B**, panels A and B of Figure 3, red) relative to that observed in THF solvent (Figure 2). Note that **PZn<sub>3</sub>** **B** and **C** display low energy absorptive and emissive bands with decreased oscillator strengths and augmented spectral breadths within PEO-b-PEE-based vesicles, relative to that observed in PEO-b-PBD-based vesicles, likely because of PEE polymer-chain-packing effects that drive an augmented PZn–PZn torsional angle distribution about a larger mean torsional angle; such a perturbation to the conformational heterogeneity disrupts **PZn<sub>3</sub>** symmetry as manifest in PBD-based bilayers (Figure 3A)<sup>28</sup> and makes more probable fluorophore aggregation within PEE membrane environments.

In comparing **PZn<sub>3</sub>** membrane incorporation in the two biodegradable vesicle systems, PEO-b-PCL (Figure 3C) and PEO-b-PMCL (Figure 3D), the **PZn<sub>3</sub>** **B** absorption and emission spectra differ substantially with respect to those determined in biocompatible PEO-b-PBD- (Figure 3A) and PEO-b-PEE (Figure 3B)-based polymersomes. In the caprolactone-based membranes, **PZn<sub>3</sub>** **B**, as well as **PZn<sub>3</sub>** **C–E**, exhibit reduced dissolution, broadened absorptive spectral manifolds, and substantially bathochromically shifted emission bands, indicative of aggregate formation.<sup>33</sup> Within all polymersome environments (Figure 3A–D), **PZn<sub>3</sub>** **A** displays uniquely conserved spectral properties that are largely identical to those that it exhibits in THF solvent (Figure 2). This is attributed to **PZn<sub>3</sub>** **A**'s relative localization at the membrane hydrophobic-block/hydrophilic-block interface, where its dissolution and dispersion are less sensitive to **PZn<sub>3</sub>**-substituent-group/hydrophobic-polymer-chain interactions that normally dictate fluorophore solvation and the nature of the local dielectric environment within the bilayer core.

**PZn<sub>3</sub> Fluorescence as a Function of Membrane-Loading Concentration.** We next compared the relative concentration-dependent fluorescence intensities of two **PZn<sub>3</sub>**-based fluorophores (**A** and **B**) within a variety of different polymersome environments. **PZn<sub>3</sub>** **A** was selected because it universally disperses within each vesicle system examined, independent of membrane chemical composition. Because of its similar 3',5'-aryl substitution pattern, **PZn<sub>3</sub>** **B** was also examined in order to help elucidate the importance of **PZn<sub>3</sub>** **A**'s 9-methoxy-1,4,7-trioxanonyl substituents, which localize **PZn<sub>3</sub>** **A** fluorophores within a smaller membrane volume element relative to those sampled by **PZn<sub>3</sub>** **B–E** emitters, in determining the steady-state integrated emission intensity as a function of membrane-loading concentration. Fluorophore membrane-loading was varied between 0.2 and 20 mol % (corresponding to vesicles formed from 500:1 and 5:1 polymer to fluorophore molar ratios, respectively). For each unique emissive polymersome formulation ( $n = 3$  experiments for each polymer:fluorophore combination), steady-state integrated-emission intensities were measured ( $\lambda_{\text{ex}} = 785$  nm,  $\lambda_{\text{em}} = 800–900$  nm), normalized by total emitter numbers in suspension (determined via a Beer's law calcula-

(33) Turro, N. J. *Modern Molecular Photochemistry*; University Science Books: Sausalito, CA, 1991.



**Figure 3.** Steady-state electronic absorption and emission (inset) spectra of  $\text{PZn}_3$ -based fluorophores A–E when incorporated within different polymersome environments. Compound color scheme corresponds to that displayed in Figure 1B. (A) Aqueous suspensions of PEO-*b*-PBD-based vesicles. (B) Aqueous suspensions of PEO-*b*-PEE-based vesicles. (C) Aqueous suspensions of PEO-*b*-PCL-based vesicles. (D) Aqueous suspensions of PEO-*b*-PMCL-based vesicles. Experimental conditions:  $T = 25\text{ }^\circ\text{C}$ , DI water,  $\lambda_{\text{ex}} = 510\text{ nm}$ .

**Table 1.** Polymersome-Forming Diblock Copolymers

polymer composition	$M_w$	$\ell$ (nm)
PEO <sub>30</sub> - <i>b</i> -PBD <sub>46</sub>	3800	9.6
PEO <sub>80</sub> - <i>b</i> -PBD <sub>125</sub>	10400	14.8
PEO <sub>40</sub> - <i>b</i> -PEE <sub>37</sub>	3900	8.0
PEO <sub>45</sub> - <i>b</i> -PCL <sub>105</sub>	14000	22.5
PEO <sub>43</sub> - <i>b</i> -PMCL <sub>66</sub>	10300	16.6

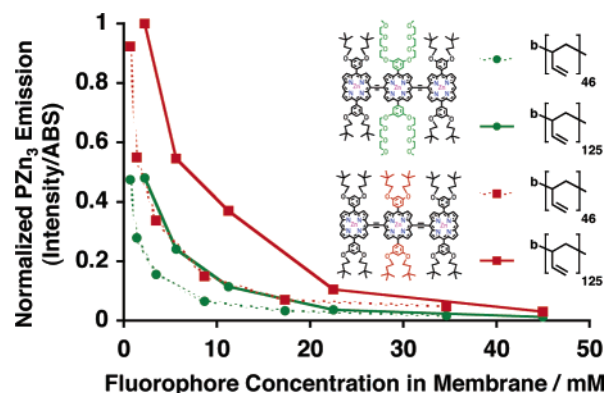
tion), and plotted on a relative scale vs the calculated concentration of emitters within the polymersome membrane.

The fluorophore concentration (mM) within the membrane's polymer matrix was calculated from the following derived expression

$$C_{\text{dye}} = \frac{1 \times 10^{27} \chi (d^2 + (d - 2\ell)^2)}{4 S_p N_A \left( \frac{3}{4} d^2 - \frac{3}{2} d\ell + \ell^2 \right)}$$

Here,  $\chi$  is the molar ratio of fluorophore/polymer,  $d$  is the vesicle's diameter ( $\sim 300\text{ nm} \pm 100\text{ nm}$  as determined by dynamic light scattering),  $S_p$  is the interfacial area per polymer chain ( $\sim 1\text{ nm}^2$  for each of the examined polymers),<sup>34</sup>  $N_A$  is Avogadro's number, and  $\ell$  is the membrane core thickness (nm). Key metrical parameters of each of the vesicle-forming diblock copolymers examined in this study are compiled in Table 1.

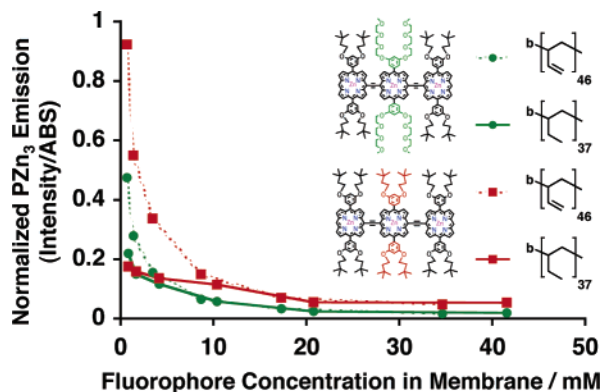
**Effects of Increasing Membrane Core Thickness of Fluorophore Dissolution.** We first examined membrane-loading of  $\text{PZn}_3$  A and B within two different poly(ethyleneoxide)-block-poly(butadiene)-based polymersomes



**Figure 4.** Relative steady-state normalized-fluorescence-emission intensities of  $\text{PZn}_3$ -based fluorophores A (green) and B (red) as a function of membrane-loading concentration, in aqueous suspensions of PEO<sub>30</sub>-*b*-PBD<sub>46</sub> (dotted) and PEO<sub>80</sub>-*b*-PBD<sub>125</sub> (solid) vesicles. Experimental conditions:  $T = 25\text{ }^\circ\text{C}$ , DI water,  $\lambda_{\text{ex}} = 785\text{ nm}$ .

to assess the effects of increasing membrane core thickness ( $\ell$ ) on relative fluorophore distribution and interfluorophore spacing. Although  $\text{PZn}_3$  A and B display nearly identical steady-state absorbance and fluorescence spectra within both PEO<sub>30</sub>-*b*-PBD<sub>46</sub> ( $\ell = 9.6\text{ nm}$ ) and PEO<sub>80</sub>-*b*-PBD<sub>125</sub> ( $\ell = 14.8\text{ nm}$ )-based vesicles, they exhibit different concentration-dependent fluorescence intensities (Figure 4). In both PEO<sub>30</sub>-*b*-PBD<sub>46</sub> (dotted) and PEO<sub>80</sub>-*b*-PBD<sub>125</sub>-based polymersomes (solid),  $\text{PZn}_3$  B (red) manifests more intense fluorescence at a given membrane concentration relative to  $\text{PZn}_3$  A (green) (Figure 4). Further, at a specific intramembranous fluorophore concentration, the effects of increasing membrane core thickness upon augmentation of integrated fluorescence intensity (comparing dotted to solid lines for

(34) Won, Y. Y.; Brannan, A. K.; Davis, H. T.; Bates, F. S. *J. Phys. Chem. B* **2002**, *106*, 3354–3364.



**Figure 5.** Relative steady-state normalized-fluorescence-emission intensities of **PZn<sub>3</sub>**-based fluorophores **A** (green) and **B** (red) as a function of membrane-loading concentration, in aqueous suspensions of PEO<sub>36</sub>-b-PBD<sub>40</sub> (dotted) and PEO<sub>40</sub>-b-PEE<sub>37</sub> (solid) vesicles. Experimental conditions:  $T = 25\text{ }^{\circ}\text{C}$ , DI water,  $\lambda_{\text{ex}} = 785\text{ nm}$ .

each fluorophore in Figure 4) are more profound for **PZn<sub>3</sub>** **B**. These results are consistent with previous observations<sup>28</sup> that hydrophobic 3,3-dimethyl-1-butyloxy groups are expected to preferentially disperse these PZn-based fluorophores into the core of the bilayer membrane (**PZn<sub>3</sub>** **B**), whereas increasing numbers of 9-methoxy-1,4,7-trioxanonyl groups should enhance the extent to which these fluorochromes are dispersed at the membrane interface of the hydrophilic PEO and hydrophobic PBD block fractions (**PZn<sub>3</sub>** **A**). Thus, increasing membrane core thickness tends to augment the available volume for **PZn<sub>3</sub>** **B** dissolution to a greater extent relative to that for the **PZn<sub>3</sub>** **A** fluorophore, thereby enlarging mean fluorophore–fluorophore intermolecular distances and membrane-localized emissive output for this fluorochrome.

**Fluorophore Loading Within Saturated and Fully Unsaturated Branched-Polymer Membranes.** We next compared vesicle loading of **PZn<sub>3</sub>** **A** and **B** within related branched poly(butadiene) and poly(ethylene)-based membranes. Diblock copolymers of PEO<sub>30</sub>-b-PBD<sub>46</sub> and PEO<sub>40</sub>-b-PEE<sub>37</sub> were utilized to generate polymersomes of similar membrane core thickness ( $\ell = 9.6$  vs 8 nm, respectively).<sup>30</sup> By comparing relative normalized emission intensities for **PZn<sub>3</sub>** **A** (green) and **B** (red) within these two related vesicle systems (Figure 5), we can assess the importance of interchain packing effects and fluorophore–fluorophore  $\pi$ – $\pi$  interactions in determining vesicular **PZn<sub>3</sub>** emissive output. Reexamination of the Figure 3 steady-state absorbance and emission spectra shows that **PZn<sub>3</sub>** **A** exhibits nearly identical spectral features within both PEO-b-PBD- (Figure 3A, green) and PEO-b-PEE-based vesicles (Figure 3B, green), whereas **PZn<sub>3</sub>** **B** (red) shows a slight decrease in its Q-state absorption manifold intensity and a relative bathochromically shifted fluorescence band  $\lambda_{\text{max}}$  within PEE ( $\lambda_{\text{max}} = 801\text{ nm}$ ) vs PBD ( $\lambda_{\text{max}} = 818\text{ nm}$ )-based membranes. Moreover, **PZn<sub>3</sub>** **B**'s relative normalized emission intensities were lower when incorporated within PEO-b-PEE (solid) relative to that observed for PEO-b-PBD (dotted)-based polymersomes (Figure 5).

Although it is difficult to isolate the etiology of these phenomena, it could potentially implicate differences in local intramembranous dielectric environments between PEE- and

PBD-based vesicles. **PZn<sub>3</sub>** **A**'s relative localization near the membrane hydrophobic-block/hydrophilic-block interface is consistent with the fact that its steady-state spectral features are unchanged when dispersed in either PEO-b-PBD- or PEO-b-PEE-based vesicles (panels A and B of Figure 3, green). At high membrane-loading concentrations ( $>5\text{ mM}$ ), **PZn<sub>3</sub>** **A** (green) exhibits nearly identical concentration-dependent emission intensities within both PEO-b-PBD- (dotted) and PEO-b-PEE-based polymersomes (solid) (Figure 5). As a result, its local dielectric environment can be assumed to be relatively unchanged and the fluorophore is likely similarly solvated when incorporated within either vesicle membrane. At lower membrane-loading concentrations ( $<5\text{ mM}$ ), however, **PZn<sub>3</sub>** **A** shows, interestingly, a significant difference in the integrated fluorescence intensity between the two membrane environments (Figure 5, green dotted vs solid).

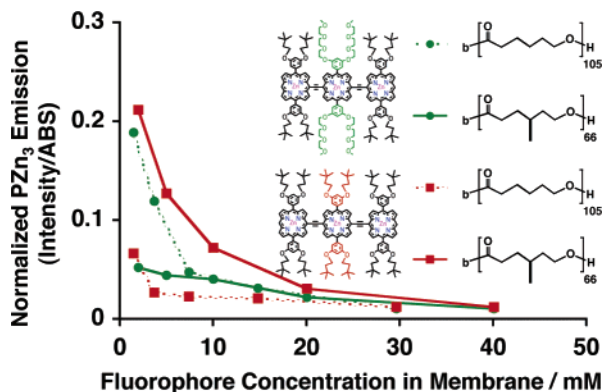
These results suggest that changes in local membrane dissolution of the **PZn<sub>3</sub>** **A** fluorophore occur with increased loading. At low membrane-loading concentrations ( $<5\text{ mM}$ ), **PZn<sub>3</sub>** **A** may be more homogeneously distributed throughout the bilayer core, in a fashion similar to that previously established for **PZn<sub>3</sub>** **B**. At higher loading concentrations ( $>5\text{ mM}$ ), although **PZn<sub>3</sub>** **A**'s emission intensities are identical within both polymersomal environments (Figure 5, green dotted and green solid lines), the integrated fluorescence is less than that observed for **PZn<sub>3</sub>** **B** (Figure 5, red dotted and red solid lines). This is consistent with other data that indicate that **PZn<sub>3</sub>** **B** is dissolved uniformly throughout the hydrophobic core at all membrane loading concentrations; in contrast, increasing membrane-dispersed **PZn<sub>3</sub>** **A** concentrations results in localization of greater numbers of fluorochromes to a more limited dissolution volume near the vesicles' aqueous/membrane interface. At low membrane loading levels, as the steady-state integrated-emission intensities are larger for both **PZn<sub>3</sub>** **A** and **B** in PEO-b-PBD relative to PEO-b-PEE-based membranes, the PBD solvation environment more effectively minimizes aggregation of emitters relative to that which exists in the more electronically insulating PEE-based bilayers.

**Fluorophore Loading Within Biodegradable Caprolactone-Based Vesicle Membranes.** We have recently described the generation of biodegradable vesicles formed through spontaneous self-assembly of pure diblock copolymers of poly(ethyleneoxide)-block-poly( $\epsilon$ -caprolactone) (PEO-b-PCL).<sup>31</sup> When compared to degradable polymersomes formed from blending “bioinert” and hydrolyzable components (where  $\tau_{1/2}$  release<sup>35,36</sup>  $\approx$   $\tau_{1/2}$  circulation<sup>37</sup>  $\sim$  tens of hours), PEO<sub>45</sub>-b-PCL<sub>105</sub>-based vesicles possess much slower release kinetics ( $\tau_{1/2}$  release  $\approx$  days), offering potential advantages for future in vivo applications. Moreover, their large membrane core thicknesses ( $22.5 \pm 2.3\text{ nm}$ ) afford the opportunity for facile incorporation of PZn-based fluorophores, enabling the generation of NIR-

(35) Ahmed, F.; Hategan, A.; Discher, D. E.; Discher, B. M. *Langmuir* **2003**, *19*, 6505–6511.

(36) Ahmed, F.; Discher, D. E. *J. Controlled Release* **2004**, *96*, 37–53.

(37) Photos, P. J.; Bacakova, L.; Discher, B.; Bates, F. S.; Discher, D. E. *J. Controlled Release* **2003**, *90*, 323–334.



**Figure 6.** Relative steady-state normalized-fluorescence-emission intensities of **PZn<sub>3</sub>**-based fluorophores **A** (green) and **B** (red) as a function of membrane-loading concentration, in aqueous suspensions of PEO<sub>45</sub>-b-PCL<sub>105</sub> (dotted) and PEO<sub>43</sub>-b-PMCL<sub>66</sub> (solid) vesicles. Experimental conditions:  $T = 25\text{ }^{\circ}\text{C}$ , DI water,  $\lambda_{\text{ex}} = 785\text{ nm}$ .

emissive polymersomes that are not only biocompatible but also fully biodegradable. Vesicles generated from pure diblock copolymers of PEO-b-PCL possess solid (crystalline-phase) membranes,<sup>31</sup> whereas those assembled from poly(ethyleneoxide)-block-poly( $\gamma$ -methyl- $\epsilon$ -caprolactone) (PEO-b-PMCL) have fluid-phase<sup>32</sup> bilayers. As such, we compared relative concentration-dependent emission intensities of **PZn<sub>3</sub>** **A** and **B** within two biodegradable polymersome systems based on PEO<sub>45</sub>-b-PCL<sub>105</sub> and PEO<sub>43</sub>-b-PMCL<sub>66</sub>, in order to evaluate how membrane crystallinity influences **PZn<sub>3</sub>** solvation and mean electronic environment (Figure 6).

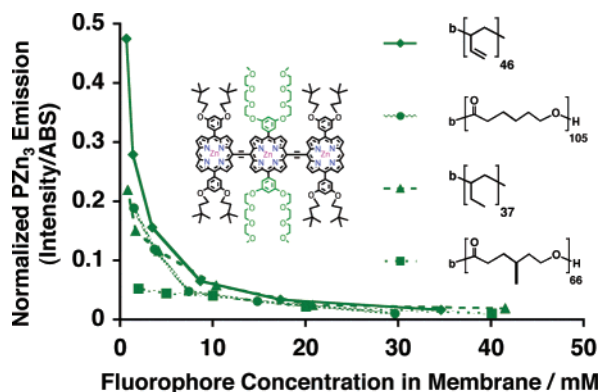
As polycaprolactone-based vesicles possess membranes that have dielectric environments that differ substantially from those derived from branched poly(ethylene) and poly(butadiene), intramembranous **PZn<sub>3</sub>** distributions for a given fluorophore need not be identical within these two classes of vesicles. **PZn<sub>3</sub>** **A**'s (green) steady-state absorption and emission spectra look identical within both PEO-b-PCL- (Figure 3C) and PEO-b-PMCL-based vesicles (Figure 3D), resembling those observed in THF and other polymersome environments. **PZn<sub>3</sub>** **B**'s spectra in both biodegradable environments (panels C and D of Figure 3, red), however, are consistent with significant intermolecular fluorophore interactions and notably possess fluorescence bands that are substantially broadened and bathochromically shifted as compared to those observed for the same fluorophore in THF or within other emissive vesicle systems (panels A and B of Figure 3). Further, at low loading levels (<5 mM), **PZn<sub>3</sub>** **A** exhibited significantly greater concentration-dependent emission intensities within crystalline PCL-based membranes (Figure 6, green dotted), relative to those derived from fluid PMCL (Figure 6, green solid). At higher membrane-loading levels (>5 mM), **PZn<sub>3</sub>** **A**'s relative concentration-dependent fluorescence was identical within both membrane environments. **PZn<sub>3</sub>** **B**'s emission, however, was uniformly greater within fluid PMCL-based vesicles (Figure 6, red solid) relative to that within PCL polymersomes (Figure 6, red dotted). Finally, although PEO<sub>45</sub>-b-PCL<sub>105</sub>-based polymersomes possess a larger membrane core thickness ( $l = 22.5\text{ nm}$ ) than those comprised of PEO<sub>43</sub>-b-PMCL<sub>66</sub> ( $l = 16.6\text{ nm}$ ),

**PZn<sub>3</sub>** displayed relatively greater fluorescence within the smaller-volume PMCL-based environment (Figure 6, red solid).

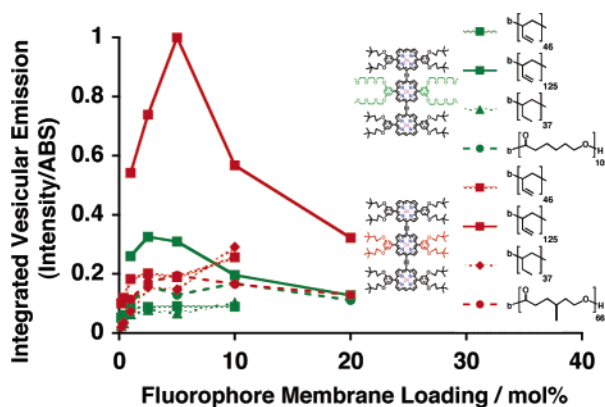
These differences in **PZn<sub>3</sub>** **A** and **B** membrane distribution and dissolution within the two different biodegradable vesicle systems are likely derived in large part from substituent-driven intermembrane interactions involving fluorophore aryl group substituents and polymer chains that dictate fluorophore solvation. Replacement of **PZn<sub>3</sub>** **B**'s central *meso*-aryl ring 3'- and 5'-3,3-dimethyl-1-butyloxy substituents with 9-methoxy-1,4,7-trioxanonyl groups (**PZn<sub>3</sub>** **A**) augments chromophore amphiphilicity and drives its dissolution into relatively more polar environments. At low fluorophore-loading levels within PCL-based membranes (<5 mM), **PZn<sub>3</sub>** **A** is likely more homogeneously distributed throughout the bilayer core, as attributed to its larger relative concentration-dependent emission intensities when compared to its dissolution within PMCL-based bilayers (Figure 6, dotted green vs red). At higher fluorophore loading levels, incorporation of **PZn<sub>3</sub>** **A** within PCL membranes resembles that within PMCL-based systems, consistent with incremental incorporation of larger numbers of fluorophores predominantly to the vesicles' aqueous/membrane interface. As compared to the more hydrophilic **PZn<sub>3</sub>** **A**, **PZn<sub>3</sub>** **B** is uniformly distributed throughout the membrane core of the relatively apolar PMCL-based membrane (Figure 6, solid red vs green lines); as a result, its concentration-dependent fluorescence is larger than that observed for **PZn<sub>3</sub>** **A** within PEO-b-PCL-based systems at all loading levels (Figure 6 red solid vs green dotted lines). Finally, **PZn<sub>3</sub>** **B**'s fluorescence is predominantly quenched within PCL-based membranes at all but the most dilute concentrations (Figure 6, dotted red line), consistent with significant aggregation of this hydrophobic fluorophore within this more polar caprolactone-based membrane environment (Figure 3C, red).

**Comparison of **PZn<sub>3</sub>**-Loading Within Various Polymersome Systems.** In considering the entire set of steady-state spectra for **PZn<sub>3</sub>** **A–E** (Figures 2 and 3), as well as the relative concentration-dependent emission intensities for **PZn<sub>3</sub>** **A** and **B** within the various polymersome systems (Figures 4–6), it is evident that the **PZn<sub>3</sub>**-based fluorophore **A** displays robust optical properties regardless of its solvation environment. Figure 7 compares **PZn<sub>3</sub>** **A**'s relative concentration-dependent fluorescence in aqueous suspensions of biocompatible PEO<sub>30</sub>-b-PBD<sub>46</sub>- and PEO<sub>40</sub>-b-PEE<sub>37</sub>- as well as biodegradable PEO<sub>45</sub>-b-PCL<sub>105</sub>- and PEO<sub>43</sub>-b-PMCL<sub>66</sub>-based polymersomes. Consistent with earlier discussion, at high intramembranous **PZn<sub>3</sub>** **A** concentrations (>10 mM), the normalized emission is nearly identical within each vesicle environment. Along with its conserved steady-state spectral features, these data indicate that the fluorophore is similarly solvated, likely located predominantly in a region near the vesicles' hydrophobic-block/hydrophilic-block interface. At lower loading levels (<10 mM), fluorescence emission is highly dependent upon the dielectric environment of the membrane matrix. Low-level, nearly concentration-independent emission (over the range of examined fluorophore loadings) is evinced in apolar PMCL-based bilayers





**Figure 7.** Relative steady-state emission intensities of **PZn<sub>3</sub>**-based fluorophore **A** as a function of membrane-loading concentration, in aqueous suspensions of PEO<sub>30</sub>-b-PBD<sub>46</sub> (diamond), PEO<sub>45</sub>-b-PCL<sub>105</sub> (circle), PEO<sub>40</sub>-b-PEE<sub>37</sub> (triangle), and PEO<sub>43</sub>-b-PMCL<sub>66</sub> (square) vesicles. Experimental conditions:  $T = 25\text{ }^{\circ}\text{C}$ , DI water,  $\lambda_{\text{ex}} = 785\text{ nm}$ .



**Figure 8.** Comparison of the relative integrated-steady-state emission of various emissive polymersome systems, as a function of mol % membrane-loading with **PZn<sub>3</sub>** **A** (green) and **B** (red). Experimental conditions:  $T = 25\text{ }^{\circ}\text{C}$ , DI water,  $\lambda_{\text{ex}} = 785\text{ nm}$ .

incorporating **PZn<sub>3</sub>** **A** (Figure 7, square). Markedly greater normalized integrated-emission intensity is observed for **PZn<sub>3</sub>** **A** dispersed in polymersomes derived from electron-rich PBD (Figure 7, diamond). Finally, its relative concentration-dependent fluorescence intensities in aqueous suspensions of PCL- (Figure 7, circle) and PEE-based vesicles (Figure 7, triangle) are similar, indicative of its incorporation within isotropic electronic environments.

**Optimization of Steady-State Fluorescence from NIR-Emissive Polymersomes.** For future biological applications, it is necessary to optimize the total-integrated vesicular-emission of polymersomes incorporating **PZn<sub>n</sub>**-based fluorophores. Choosing the appropriate polymer:fluorophore molar loading level, which dictates the effective intramembranous fluorophore concentrations and hence mean intermolecular spacing, involves a tradeoff between larger relative emission per fluorophore and greater fluorophore copy numbers per vesicle. Figure 8 displays the total integrated vesicular emission of **PZn<sub>3</sub>** **A**- and **PZn<sub>3</sub>** **B**-based biocompatible and biodegradable emissive polymersomes. Emission of PEO<sub>80</sub>-b-PBD<sub>125</sub>-based vesicles incorporating **PZn<sub>3</sub>** **B** (Figure 8, solid red line with squares) is greatest and maximized at an approximately 2.5 mol % loading level. In general, total integrated emission intensities of polymersomes incorporating the more hydrophobic **PZn<sub>3</sub>**-based fluorophore **B** (Figure 8,

red) are greater than identical vesicles incorporating **PZn<sub>3</sub>** **A** (Figure 8, green). Further, among vesicles incorporating the same **PZn<sub>3</sub>** fluorophore (either **A** or **B**), emissive output tends to correlate with membrane core thickness: emission from PEO<sub>80</sub>-b-PBD<sub>125</sub>-based polymersomes (Figure 8, solid lines with square) is greater than those vesicles comprised of PEO<sub>30</sub>-b-PBD<sub>46</sub> (Figure 8, dotted lines with square) or PEO<sub>40</sub>-b-PEE<sub>37</sub> diblock copolymers (Figure 8, dotted lines with diamond), which possess thinner bilayers. Finally, the total vesicular emissive outputs from PEO<sub>45</sub>-b-PCL<sub>105</sub>-based polymersomes incorporating **PZn<sub>3</sub>** **A** (Figure 8, dotted green line with circle) are nearly identical to those of PEO<sub>43</sub>-b-PMCL<sub>66</sub>-based vesicles dispersing **PZn<sub>3</sub>** **B** (Figure 8, dotted red line with circle), because of appropriately matched fluorophore substituents driving polymer-membrane solvation; consistent with these results, **PZn<sub>3</sub>** **A** displays a low-level of concentration-independent emission within apolar PMCL-based membranes (Figure 6, solid green line), whereas hydrophobic **PZn<sub>3</sub>** **B** displays similar aggregate-like behavior (Figure 6, dotted red line) within more polar PCL-derived polymersomes.

## Conclusions

We have demonstrated that the nature of intramembranous polymer-fluorophore physicochemical interactions profoundly influences the bulk optical properties of NIR-emissive polymersomes. Specifically, appropriate selection of fluorophore ancillary aryl group substituents and choice of polymer chain chemistries modulate fluorophore solvation, mean electronic environment, and concentration-dependent fluorescence. **PZn<sub>3</sub>**-based fluorophores with 3',5'-disubstituted pendent aryl groups (**PZn<sub>3</sub>** **A** and **B**) are more effectively membrane-dispersed than identical fluorophores that possess a 2',6'-substitution pattern (**PZn<sub>3</sub>** **C**) or those that lack 10',20'-phenyl rings (**PZn<sub>3</sub>** **D** and **E**). Further, appropriate selection of ancillary aryl group substituents drives fluorophore solvation into membranous regions with matching dielectric properties.

Within biocompatible PEO-b-PBD- and PEO-b-PEE-based polymersomes, hydrophobic **PZn<sub>3</sub>** **B** exhibited the highest per molecule fluorescence at all membrane-loading levels. Emissive output was further augmented within thicker vesicle membranes comprised of the more electron-rich PBD-based matrix, relative to that manifest in polymersomes featuring PEE-derived hydrophobic block fractions. **PZn<sub>3</sub>** emission within the two biodegradable vesicle systems, PEO-b-PCL and PEO-b-PMCL, was also similarly incumbent on specific ancillary aryl group substitution. Hydrophobic **PZn<sub>3</sub>** **B** exhibited larger normalized emission intensities within the relatively apolar PMCL-based membrane environment, whereas **PZn<sub>3</sub>** **A**'s increased amphiphilicity enabled its effective dissolution within more polar PCL-based bilayers.

The optimization of NIR-emissive polymersomes requires: (1) elucidation of fluorophore photophysics in order to design emitters with appropriately large NIR extinction coefficients and fluorescence quantum yields, (2) the regulation of the average fluorophore-fluorophore interspatial separations in order to facilitate fluorophore emission, and

(3) the loading of appropriate numbers of fluorophores within a single vesicle composite in order to maximize signal sensitivity. Moreover, incorporation of **PZn<sub>n</sub>**-based near-infrared fluorophores within poly(caprolactone)-based membranes heralds the introduction of NIR-emissive polymerosomes that are not only biocompatible, but now also fully biodegradable. Development of these new self-assembled, colloidal nanoparticles offers insights into the design of other high-irradiance, nanoscale organic probes for in vivo optical imaging applications.

**Acknowledgment.** This work was supported by a grant from The National Institutes of Health (R01CA115229). M.J.T. and D.A.H. thank the MRSEC Program of the National Science Foundation (DMR05-20020) for infrastructural support. D.A.H. also thanks the National Institutes of Health (EB003457-01), and P.P.G. acknowledges fellowship support from the NIH Medical Scientist Training Program and the Whitaker Foundation.

CM062427W